

The $\text{Cu}_2\text{S}-\text{Cu}_7\text{PS}_6-\text{Cu}_6\text{PS}_5\text{I}$ quasiternary system

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The $\text{Cu}_2\text{S}-\text{Cu}_7\text{PS}_6-\text{Cu}_6\text{PS}_5\text{I}$ quasiternary system was investigated by differential thermal analysis and X-ray diffraction in combination with mathematical modeling. The phase diagram of the $\text{Cu}_7\text{PS}_6-\text{Cu}_6\text{PS}_5\text{I}$ system, perspective views of the phase interactions in the ternary system, and a projection of the liquidus surface were constructed. New complex compounds do not form. The ternary system is of the invariant eutectic type and characterized by the formation of limited solid solutions. The three-dimensional one-, two- and three-phase regions present in the $\text{Cu}_2\text{S}-\text{Cu}_7\text{PS}_6-\text{Cu}_6\text{PS}_5\text{I}$ system are described.

Phase diagram / Thermal analysis / X-ray diffraction / Superionic conductors

Introduction

The large family of chalcogenide and halide-chalcogenide compounds called argyrodites (from the name of the mineral argyrodite, Ag_8GeS_6) [1] has been investigated for a long time, due to interesting physical and chemical properties: phase transitions at low temperatures, nonlinear ferroelectric properties, high ionic conductivity. Superionic conductivity in argyrodites is associated with Ag^+ and Cu^+ [2-4] in the cation sublattice. The $\text{Cu}_2\text{S}-\text{CuI}-\text{P}_2\text{S}_5$ quasiternary system is interesting for the investigation of the presence of ternary and quaternary compounds belonging to the argyrodite family, which exhibit superionic properties and have practical application. Triangulation of $\text{Cu}_2\text{S}-\text{CuI}-\text{P}_2\text{S}_5$ quasiternary system [5] divides it into four secondary quasiternary systems. The phase equilibria in the $\text{Cu}_2\text{S}-\text{CuI}-\text{Cu}_6\text{PS}_5\text{I}$ secondary quasiternary system have been studied previously [6]. Investigation of the $\text{Cu}_2\text{S}-\text{Cu}_7\text{PS}_6-\text{Cu}_6\text{PS}_5\text{I}$ secondary quasiternary system is a necessary step in the study of the $\text{Cu}_2\text{S}-\text{CuI}-\text{P}_2\text{S}_5$ quasiternary system, which may be characterized by the formation of complex compounds and solid solutions based on them.

Experimental

Synthesis of initial Cu_2S and CuI was carried out with high-purity elements (better than 99.99 wt.%). The binary compounds were prepared by the single- and two-temperature method from stoichiometric amounts of the initial elements in evacuated quartz containers, and purified by the zone crystallization method. The

Cu_7PS_6 and $\text{Cu}_6\text{PS}_5\text{I}$ ternary compounds and multicomponent alloys were synthesized from binary Cu_2S and CuI and elementary phosphorous and sulfur in quartz ampoules evacuated to a residual pressure of 0.13 Pa. The highest synthesis temperature was 923 K. After thermal treatment at the highest temperature for 120 h, the samples were slowly cooled (25-30 K per hour) to 573 K and homogenized at this temperature for 336 h. Subsequently the ampoules were quenched in cold water. The samples were heated and cooled in a furnace equipped with an RIF-101 programmer, which provided a linear temperature variation. The phase equilibria in the ternary system were studied by classical methods of physico-chemical analysis, such as differential thermal analysis (Pt-Pt/Rh thermocouple with an accuracy of ± 5 K) and X-ray powder diffraction (DRON-3, $\text{Cu } K_\alpha$ radiation, Ni filter), in combination with the simplex method of mathematical modeling of phase equilibria in multicomponent systems [7]. This method provides good results and allows reducing the number of alloys synthesized in the ternary system.

Results and discussion

Based on the DTA and XRD results, the phase diagram of the $\text{Cu}_7\text{PS}_6-\text{Cu}_6\text{PS}_5\text{I}$ system was constructed. It belongs to the Roozeboom type V (Fig. 1). It contains γ - and γ' -solid solution ranges of the low- (lt) and high-temperature (ht) modifications of $\text{Cu}_6\text{PS}_5\text{I}$, and β - and β' -solid solution ranges of lt- and ht- Cu_7PS_6 , respectively. There exists a eutectic point (invariant process $\text{L} \leftrightarrow \beta' + \gamma'$) in the system with the coordinates 85 mol.% Cu_7PS_6 , 1254 K. The extent

of the β'-solid solution range at the eutectic temperature is 65 mol.% Cu₆PS₅I, that of the γ'-solid solution range is 8 mol.% Cu₆PS₅I. The peritectoid process on the basis of the polymorphous interaction of the Cu₇PS₆ ternary compound γ'+β'↔γ takes place at 751 K (the coordinate of the peritectoid point is 6 mol.% Cu₆PS₅I). The peritectoid process on the basis of the polymorphous interaction of the Cu₆PS₅I ternary compound γ'+β'↔β takes place at 756 K (the coordinate of the peritectoid point is 92 mol.% Cu₆PS₅I). The extent of the γ-solid solution range at 573 K is 15 mol.% Cu₇PS₆. At the same temperature the β-solid solution extends to 8 mol.% Cu₆PS₅I.

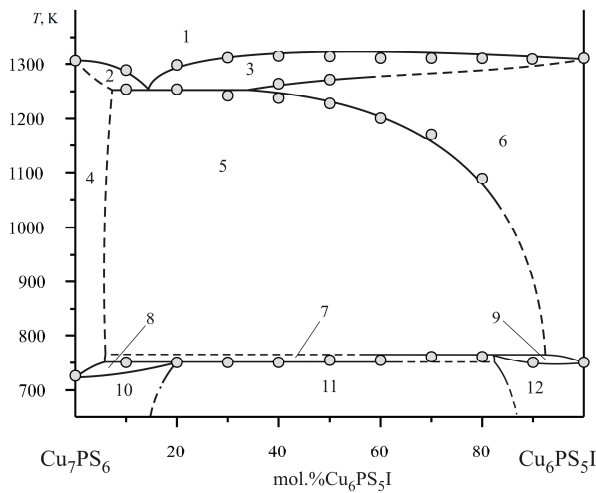


Fig. 1 Phase diagram of the Cu₇PS₆–Cu₆PS₅I system: (1) L, (2) L+γ', (3) L+β', (4) γ', (5) γ'+β', (6) β', (7) γ'+β, (8) γ+γ', (9) β+β', (10) γ, (11) γ+β, (12) β.

Based on the results of the investigation by DTA and the simplex method, a projection of the Cu₂S–Cu₇PS₆–Cu₆PS₅I liquidus surface onto the concentration triangle was constructed (Fig. 2). The liquidus of the system consists of three primary crystallization areas: the α'-phase (bordered by Cu₂S–e1–E–e2–Cu₂S), β'-phase (bordered by Cu₆PS₅I–e1–E–e3–Cu₆PS₅I) and γ'-phase (bordered by Cu₇PS₆–

e2–E–e3–Cu₇PS₆). The fields of primary crystallization are divided by three monovariant lines: e1–E (temperature range 1233–1219 K), e2–E (temperature range 1260–1219 K), e3–E (temperature range 1254–1219 K), which cross at the ternary invariant eutectic point E (L↔α'+β'+γ').

The lines of the monovariant equilibria were described using polynomial analyses of vertical sections. For example, the results of the analysis of the section a1–a1' (5 mol.% Cu₂S) are shown in Fig. 3 and Table 1. The primary crystallization of the β'- and γ'-phases is described by a polynomial $Y = a_0 + a_1X + a_2X^2 + a_3X^3$ (Y represents the temperature of crystallization in K, X is the concentration in mol.%, and a₀, a₁, a₂, a₃ are polynomial coefficients). The polynomial curves cross at a point situated on the e3–E monovariant line of the Cu₂S–Cu₇PS₆–Cu₆PS₅I quasiternary system. It should be noted that the results of the calculations showed good agreement with the experimental data (correlation parameter r_{xy} 0.9963–0.9999; root mean square deviation 0.2963–5.1976).

The coordinates of the invariant eutectic point (E) was determined by computer modeling using the simplex method: 60 mol.% Cu₂S, 34 mol.% Cu₇PS₆, 6 mol.% Cu₆PS₅I, 1219 K.

A perspective view of the Cu₂S–Cu₇PS₆–Cu₆PS₅I quasiternary system is shown in Fig. 4. The points A'', B'' and C'', which are located at the edges of a ternary prism, represent the melting temperature of the three components (Cu₆PS₅I – 1311 K, Cu₂S – 1396 K, Cu₇PS₆ – 1318 K). Limited solid solutions form in the system: α- and α'-phases based on the lt- and ht-modifications of Cu₂S, β- and β'-phases based on the lt- and ht-modifications of Cu₆PS₅I, γ- and γ'-phases on the lt- and ht-modifications of Cu₇PS₆. The faces of the ternary prism belong to the three binary eutectic type systems Cu₂S–Cu₆PS₅I, Cu₂S–Cu₇PS₆ and Cu₇PS₆–Cu₆PS₅I, which are characterized by the invariant eutectic processes L↔α'+β' (coordinates of the eutectic point e1: 20 mol.% Cu₆PS₅I, 1233 K), L↔α'+γ' (coordinates of the eutectic point e2: 50 mol.% Cu₇PS₆, 1260 K), and L↔β'+γ' (coordinates of the eutectic point e3: 85 mol.% Cu₇PS₆, 1254 K), respectively.

Table 1 Results of polynomial calculations for the a1–a1' section (10 mol.% Cu₂S).

| Fields of primary crystallization | | | | | |
|--|------|--|---|------|--|
| β'-phase (based on ht-Cu ₆ PS ₅ I) | | | γ'-phase (based on ht-Cu ₇ PS ₆) | | |
| mol.% a1' | T, K | Polynomial parameters | mol.% a1' | T, K | Polynomial parameters |
| 0 | 1305 | a ₀ = 1304.60 | 85.6 | 1250 | a ₀ = -1666.19 |
| 15.2 | 1302 | a ₁ = 7.81×10 ⁻² | 88.6 | 1275 | a ₁ = 58.82 |
| 36.4 | 1295 | a ₂ = -1.17×10 ⁻² | 92.4 | 1300 | a ₂ = -0.29 |
| 54.5 | 1278 | a ₃ = 2.91×10 ⁻⁵ | 100 | 1324 | |
| 81.1 | 1250 | r _{xy} = 0.9988 S _{xy} = 2.2534 | | | r _{xy} = 0.9999 S _{xy} = 0.2962 |

Point of intersection: 84.8 mol.% a1'; T = 1245 K.

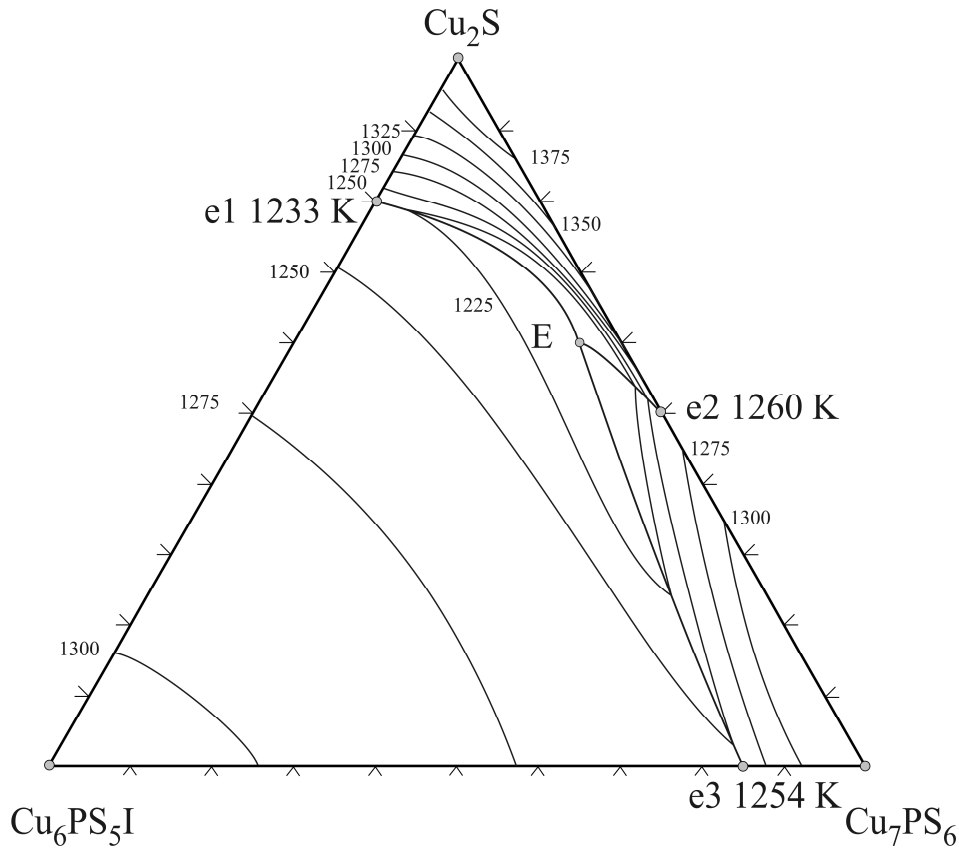


Fig. 2 Liquidus projection of the Cu_2S - Cu_7PS_6 - $\text{Cu}_6\text{PS}_5\text{I}$ quasiternary system.

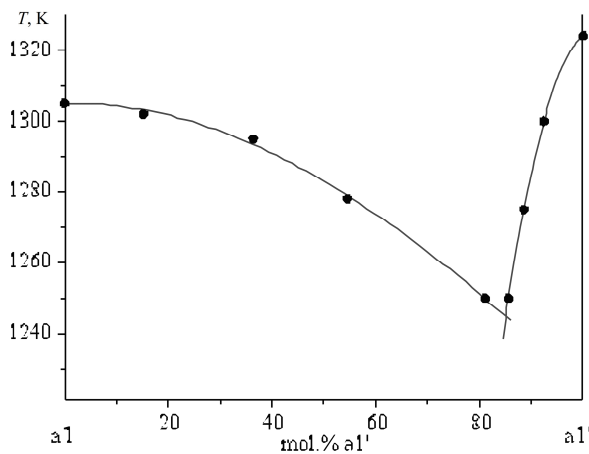


Fig. 3 Polynomial curves of primary crystallization of the β' - and γ' -phases (vertical section along the $a1$ - $a1'$ section, 10 mol.% Cu_2S).

The system liquidus is characterized by the following surfaces of primary crystallization: α' -phase (bordered by B'' - $e1$ - E - $e2$ - B''), β' -phase (bordered by A'' - $e1$ - E - $e3$ - A'') and γ' -phase (bordered by

C'' - $e2$ - E - $e3$ - C''). The lines of the monovariant processes ($e1$ - E , $e2$ - E and $e3$ - E) intersect in the invariant eutectic point E ($L \leftrightarrow \alpha' + \beta' + \gamma'$). The system solidus consists of the following fields: A'' - $a1$ - $a11$ - $a6$ - A'' (end of β' -phase crystallization), B'' - $b1$ - $b11$ - $b6$ - B'' (end of α' -phase crystallization), C'' - $c1$ - $c11$ - $c6$ - C'' (end of γ' -phase crystallization), $a1$ - $b1$ - $b11$ - $a11$ - $a1$ (end of crystallization of $\alpha' + \gamma'$), $a6$ - $a11$ - $b11$ - $b6$ - $a6$ (end of crystallization of both $\alpha' + \beta'$), $b6$ - $c1$ - $c11$ - $b11$ - $b6$ (end of crystallization of $\beta' + \gamma'$), and a triangular surface of ternary eutectic processes $a11$ - $b11$ - $c11$ - $a11$. In the subsolidus area, three invariant surfaces represent the polymorphous interactions of Cu_2S ($b14$ - $a14$ - $c14$ - $b14$) at 628 K, Cu_7PS_6 ($c13$ - $a13$ - $b13$ - $c13$) at 781 K, and $\text{Cu}_6\text{PS}_5\text{I}$ ($a12$ - $b12$ - $c12$ - $a12$) at 803 K.

The type, temperature and coordinates of the invariant processes in the Cu_2S - Cu_7PS_6 - $\text{Cu}_6\text{PS}_5\text{I}$ quasiternary system are listed in **Table 2**.

The X-ray analysis of Cu_2S - Cu_7PS_6 - $\text{Cu}_6\text{PS}_5\text{I}$ alloys of the isothermal section at 573 K confirmed that the extent of the homogeneity regions of the β - and γ -phases corresponds to 8-15 mol.%, whereas that of the α -phase is not more than 5 mol.%. New complex compounds were not observed in the ternary system.

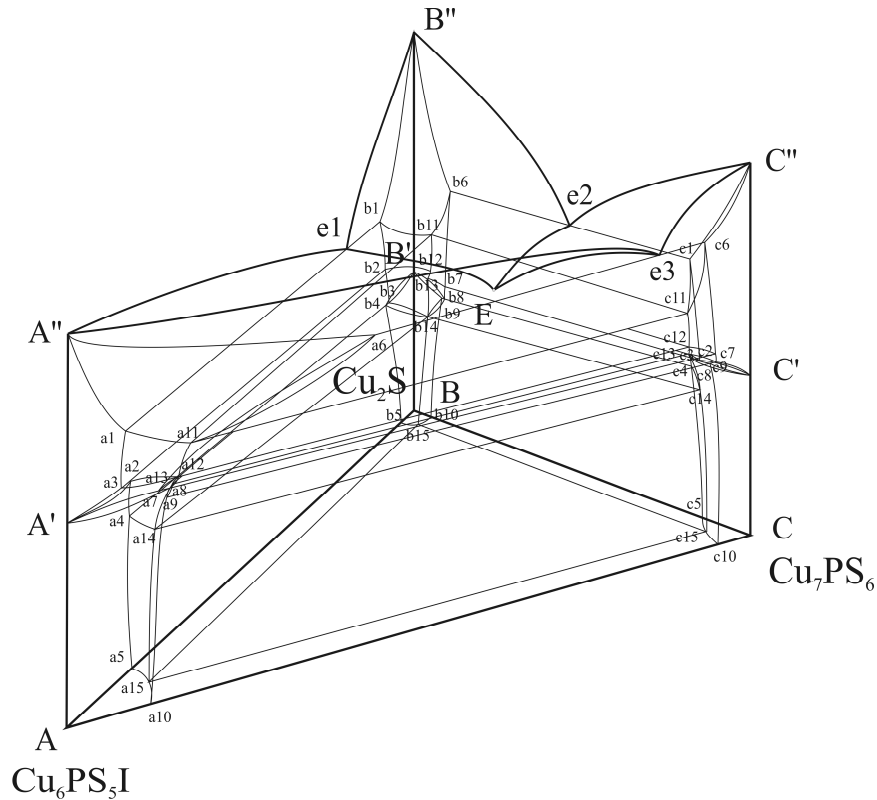


Fig. 4 Perspective view of the Cu_2S – Cu_7PS_6 – $\text{Cu}_6\text{PS}_5\text{I}$ quasiternary system.

Table 2 Type, temperature and coordinates of the invariant processes in the Cu_2S – Cu_7PS_6 – $\text{Cu}_6\text{PS}_5\text{I}$ quasiternary system.

| Type | Process | T , K | Coordinates, mol. % | | |
|----------------------------|---|---------|-----------------------|--------------------------|----------------------------------|
| | | | Cu_2S | Cu_7PS_6 | $\text{Cu}_6\text{PS}_5\text{I}$ |
| Binary invariant eutectic | e1 ($\text{L} \leftrightarrow \alpha' + \beta'$) | 1233 | 80 | – | 20 |
| Binary invariant eutectic | e2 ($\text{L} \leftrightarrow \alpha' + \gamma'$) | 1260 | 50 | 50 | – |
| Binary invariant eutectic | e3 ($\text{L} \leftrightarrow \beta' + \gamma'$) | 1254 | – | 85 | 15 |
| Ternary invariant eutectic | E ($\text{L} \leftrightarrow \alpha' + \beta' + \gamma'$) | 1219 | 60 | 34 | 6 |

Conclusions

Differential thermal analysis, X-ray diffraction and mathematical modeling of the phase equilibria in the multicomponent systems by the simplex method were used to construct for the first time the phase diagram of the Cu_7PS_6 – $\text{Cu}_6\text{PS}_5\text{I}$ system, a perspective representation of the Cu_2S – Cu_7PS_6 – $\text{Cu}_6\text{PS}_5\text{I}$ quasiternary system and a projection of the liquidus surface onto the concentration triangle. The character of the monovariant processes, and the temperatures and coordinates of the invariant processes in the ternary system were determined. The system is characterized by the invariant eutectic process $\text{L} \leftrightarrow \alpha' + \beta' + \gamma'$ (60 mol.% Cu_2S , 34 mol.% Cu_7PS_6 , 6 mol.% $\text{Cu}_6\text{PS}_5\text{I}$, 1219 K). The existence of solid solutions of the Cu_2S , Cu_7PS_6 , and $\text{Cu}_6\text{PS}_5\text{I}$ compounds was established. New complex compounds were not observed in the ternary system.

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